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Nanoscale surface modifications of medically-relevant metals: state-of-the art and perspectives

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Abstract

Evidence that nanoscale surface properties stimulate and guide various molecular and biological processes at the implant/tissue interface is fostering a new trend in designing implantable metals. Cutting-edge expertise and techniques drawn from widely separated fields, such as nanotechnology, materials engineering and biology, have been advantageously exploited to nanoengineer surfaces in ways that control and direct these processes in predictable manners. In this review, we present and discuss the state-of-the-art of nanotechnology-based approaches currently used to modify the surface of metals used for orthopedic and dental applications, and also briefly consider their use in the cardiovascular field. The effects of nanoengineered surfaces on various *in vitro* molecular and cellular events are firstly discussed. Importantly, this review also provides an overview of *in vivo* and clinical studies with nanostructured metallic implants, and addresses the potential influence of nanotopography on biomechanical events at interfaces. Ultimately the objective of this work is to give the readership a comprehensive picture of the current advances, future developments and challenges in the application of the infinitesimally small to biomedical surface science. We believe that an integrated understanding of the *in vitro* and particularly of the *in vivo* behavior is mandatory for the proper exploitation of nanostructured implantable metals and, as a matter of fact, all biomaterials.

1. Introduction

Current generations of dental and orthopedic implants are relatively effective but they still need significant improvement, particularly in their capacity to selectively influence and guide cell and tissue events at the implantation site. Their longevity is limited, and their success depends on the patient's overall health. The benefit of better mastication offered by implant-supported prostheses not only improves the quality of life but also has an impact on the overall health of the patient. Dental implants are generally limited to patients with "good

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bone quality”, leaving out a large segment of the population that has lost supporting jaw bone with age. In orthopedics, there is an increasing number of cases of total hip replacement to treat advanced degenerative changes (such as those caused by *arthritis* and *osteonecrosis*), as well as fractures in patients with osteoporosis. Here, also overall bone status plays a major role in securing both short- and long-term implant stability. The growing frequency of joint replacements, not only in a population that lives longer but also in younger people, has led to an increasing number of patients requiring revision surgery, with the average implant needing replacement approximately every 10 years.¹ The need for revision surgery typically arises from loss of bone supporting the prosthesis (*osteolysis*). Because of the additional bone loss and restructuring at the surgical site, revisions are more difficult to perform and generally result in less support for the prosthesis and poorer overall outcomes than the original surgery. Most importantly for the patient, they are associated with higher morbidity. In addition, they are extremely costly for health providers and a significant drain on healthcare resources in terms of surgeon time and post-surgery care of patients. Hip replacements are familiar examples, but there are similar needs for improved procedures for knee and spine implants.

Similarly, each year millions of individuals undergo balloon angioplasty to unblock arteries. In the U.S., about 80% of them also receive a self-expanding, metal mesh tube, called a stent, intended to prevent renarrowing of blood vessels after they have been reopened.² Balloon angioplasty has been performed since the late 1970's, but coronary stents were only introduced in the early 1990's. Major problems associated with stents are thrombosis and closure of the blood vessel after surgery (*restenosis*), the latter occurring in up to 30% of patients.² A promising development in stent technology is the advent of drug-eluting stents (DESs), whose surface is coated with polymers containing agents that regulate cell division and prevent clot formation. So far, clinical results with DESs are encouraging; however, there are still problems with this technology. For instance, the polymer coating may weakly adhere, cause inflammation and increase the propensity for thrombosis.³

Different classes of materials (metals, ceramics, polymers and composites) are currently used to manufacture prosthetic implants and biomedical devices.⁴ However, metals still represent the gold standard in implantology because of their mechanical properties, resistance to corrosion and biocompatibility,^{4, 5} which can be defined as the ability of a material to accomplish specific biomedical functions without causing adverse immune and tissue reactions. Stainless steels (such as 316L) as well as titanium and its alloys (such as TiAl and shape-memory TiNi alloys) are widely encountered in orthopedics, dentistry and cardiology.^{5–7} In addition, because of their low friction coefficient and high wear resistance, CrCo alloys are also currently used to manufacture components of knee and hip joints.^{8, 9} For implantation in the spine, tantalum has been exploited to create three-dimensional porous structures that favor bone ingrowth for a more effective osseointegration.¹⁰

Degradable metallic biomaterials (DMMs), such as magnesium- and iron-based alloys, are increasingly gaining interest for skeletal and cardiovascular applications (e.g. wires, fixation plates, stents) which do not require the permanent presence of an implanted device.¹¹ They essentially have the capacity to provide specific functions (e.g. structural support, favoring the healing process) and dissolve through corrosion afterwards, without generating toxic by products.^{12, 13}

Progress in nanotechnology now makes it possible to precisely design and modulate at the nanoscale the surface properties of materials used for various applications in medicine, offering new prospects for the patient.^{14, 15} Nanoengineered surfaces possess the unique capacity of directly affecting the molecular and cellular events that ultimately determine the overall biological response to an implanted material, such as protein adsorption, cell

adhesion and proliferation among others.^{16–20} As a result of this exceptional ability, various nanotechnology-based techniques have been developed to generate nanoscale surface features on existing biocompatible materials (reviewed in references 14, 15).

Due to their extensive use in medicine and to their potential for future biomedical applications, metals have been given a particular attention (reviewed in reference 21). Therefore, in this review paper, we bring attention to methods currently used for nanostructuring the surfaces of implantable metals, highlighting in particular those approaches that offer a clinical applicability or that have already resulted in commercial devices in the orthopedic and dental fields. We then provide evidence of the enhanced biological activity achieved with nanotechnology-based surface modifications by presenting an overview of *in vitro* studies on how cells respond to nanostructured metallic surfaces. In this context, unlocking the precise mechanisms which govern cell-substrate interactions will ultimately permit to endow implantable metals with the exact physicochemical properties needed to elicit a specific biological outcome (e. g. osseointegration, anticoagulation, etc). However, although essential in the progress of biological surface science and in the development of the future generation of implantable materials, *in vitro* experiments, as it is well recognized in biomedical sciences, only reflect part of the multifactorial and dynamic biological environment of living organisms. This translates into the need to ultimately complete the study and evaluate the performance of a biomaterial *in vivo*, an aspect that is often forgotten or just briefly discussed in review articles on nanostructured materials. Such oversight must be addressed, and this work is meant to take a step in this direction by illustrating not only the *in vitro* outcomes, but by also integrating the *in vivo* biological and biomechanical performances of nanoengineered implants. We ultimately bring attention to some important considerations we believe need to be made to correlate *in vitro* conditions with the *in vivo* reality, concluding with our perspectives for the future development of improved metals for biomedical applications.

2. Approaches for nanoscale surface modification

The surface properties of implantable metals can be modified on a range of scales by various techniques.^{22, 23} Various approaches have been used so far to create micron-scale topographies on the surface of biocompatible metals.^{24–26} Although the resulting surface features demonstrated to be effective in enhancing *in vitro*^{27–29} and *in vivo*^{30–33} biological events, it is now recognized that material-host tissue interactions are principally governed by nanometric surface cues.^{14, 16, 17, 34, 35} As a consequence, micron-scale features can only at best have an indirect influence on cellular activity and thereby can inherently only have limited activity and success. Therefore, there is a need for surface features that can have a more direct and rapid outcome. To this end, various strategies have been devised and implemented to nanoengineer surfaces that can directly influence the biological functionalities of implantable metals.^{14, 21} Only some of these methods, however, can at this time be easily exported to a large-scale production for medical implant manufacturing. In this section, we therefore present the approaches that have the potential for industrial exploitation. These techniques, divided into chemical and physical methods, have been selectively chosen according to the following parameters, necessary for large-scale manufacturing: (1) ability to simultaneously reach all surfaces in devices with complex geometries (e. g. femoral stems, dental screws and cardiovascular stents); (2) possibility to modify at the nanoscale commercially-available biocompatible metals and implants; and (3) simple integration in the industrial process line.

2.1 Chemical methods

Electrochemical modification is one of the most common and flexible ways to modify metallic surfaces on the nanoscale.³⁶ Anodic oxidation has been successfully used to

transform smooth titanium surfaces into nanotubular structures with diameters inferior to 100 nm (Figure 1).^{37–40} By adjusting parameters such as the chemistry of the electrolyte, voltage and current density, one can control physicochemical properties of surfaces,^{41, 42} and the diameter and the spacing between nanotubes can be precisely modulated.^{43, 44} In addition, by adjusting the applied potential, it is also possible to transform the protective amorphous oxide layer into one of its crystalline forms.⁴⁵ On titanium surfaces, anodization also permits to create, through a porous alumina mask, pillar-like nanostructures with tunable sizes as well as to deposit 10 μm -long titania nanotube arrays.^{46, 47} Nanostructured layers on various metallic surfaces have been similarly created using electrophoretic deposition.³⁶ For example, nanocrystalline hydroxyapatite (nano-HA) coatings (crystals size in the 15–25 nm range) and multi-walled carbon nanotubes have been deposited on titanium-based metals, resulting in an improved bioactivity.^{48, 49}

A simple chemical patterning approach using combinations of strong acids (or bases) and oxidants has been shown to efficiently generate networks of nanopits (pit diameter ranging from 20 to 100 nm) on titanium, Ti6Al4V and CrCoMo alloys, and tantalum (Figure 2).^{50, 51} Surface topography, wettability, micro and nanoroughness, as well as the thickness of the protective oxide layer, can be precisely controlled by adjusting the length of exposure, the temperature and the composition of the etching solutions.^{51–53} In addition, varying the nature of the etching solution makes it possible to incorporate selected elements (e.g. fluorine, which has antibacterial effects⁵⁴ and contributes to bone formation⁵⁵) in nanotopographic surfaces created by oxidative treatment.⁵⁰

Anodic oxidation and chemical etching have been combined to create metal/polymer composites with enhanced biological properties. Anodized nanotubular titanium has been coated with NaOH-treated nanoporous poly-lactic-co-glycolic acid (PLGA). The generated nanostructured Ti/PLGA composite stimulated cell activity, but compared to anodized titanium, no significant difference was seen.⁵⁶ A one-step chemical method based on a combination of NaOH and hydrothermal treatments has also been applied to titanium to create a wide variety of bioactive nanostructures, such as nanoleaves, nanoneedles, nanorods, nanotubes and multi-scale octahedral whiskers (Figure 3).⁵⁷

The properties of metallic surfaces can also be modified at the nanometric level by sol–gel chemistry and chemical vapor deposition (CVD), among others.^{58, 59} Niobium oxide and diamond-like carbon presenting characteristic nanotopographies have been deposited by these techniques on titanium and other substrates, providing additional avenues for improving the bioactivity of implantable metals.^{60–63}

A different strategy to improve commonly-used implantable metals is grafting bioactive molecules by simple adsorption or covalent linkage, usually peptides or proteins, to reproduce the biochemical environment that naturally sustains biological processes in the body (reviewed in references 64–66). These are usually peptides which regulate cellular adhesion (e.g. RGD-containing peptides) or extracellular matrix (ECM) proteins (e.g. collagen type I, fibronectin, vitronectin, bone sialoprotein) that coordinate the mineralization process.^{67–70} Growth factors such as bone morphogenic proteins (BMPs) and transforming growth factor beta (TGF β) participate in the recruitment and final osteogenic differentiation of undifferentiated mesenchymal cells.^{66, 71–73} In addition, pectic polysaccharides, a group of molecules normally present in plants, have demonstrated the capacity to activate cell proliferation.⁷⁴ Chitosan, a molecule generally extracted from crustaceans, has also been used to coat titanium and other metals.^{75, 76} Coatings with such molecules thus represent additional strategies for improving the biocompatibility of medical devices. The same complexing approach can be exploited to graft antibacterial agents (e.g. lysozyme and/or

poly(ethylene glycol)⁷⁷ and antibiotics,^{78–82} as well as more complex structures such as self-assembled organic helical rosette nanotubes.^{83, 84}

Most works so far have dealt with the simple adsorption of molecules. However, silanes and phosphonates have shown to be very effective spacers to covalently link bioactive proteins and peptides to titanium surfaces.^{79, 85–87} Noteworthy, silanes and phosphonates have the additional capacity to directly affect cellular functions.^{88, 89}

Molecular coatings have not been developed solely to favor and enhance cellular functions. In fact, in addition to antibacterial surfaces, there are other applications where the capacity to prevent adhesion is strongly required. For example, in angioplasty, the adhesion of platelets and uncontrolled cell growth onto the inner surface of cardiovascular stents may potentially result in the obstruction of the vessel. For this reason, biochemical surface coatings composed of non-fouling^{90, 91} and anticoagulant⁹² agents have been created to suppress interactions between the metallic surface and the surrounding biological environment. Interestingly, a common protein such as bovine serum albumin (BSA) can be used to control the subsequent adsorption cascade by exploiting its capacity to efficiently block non-specific interactions.⁹³ In the case of cardiovascular stents, biochemical functionalization has been applied to deposit plasmid DNA and self-assembled monolayers (SAMs) that provide drug-eluting capacities.^{94, 95} Finally, electrodeposition of nanoparticles and their slow-release has recently been reported as an alternative for the targeting cardiovascular malfunctions.⁹⁶

2.2 Physical methods

A variety of physical methods has been used to generate bioactive nanotopography on metal surfaces. These include plasma⁹⁷ and physical vapor deposition (PVD).^{98, 99} In addition, self-assembled nanonodules and nanorough tantalum layers with well-controlled nanoroughness have been deposited on titanium by e-beam physical vapor deposition technology.^{99, 100}

A particular category of physical methods includes technologies which provoke atomic rearrangements, such as ion implantation and thermal oxidation. Approaches based on ion implantation (reviewed in reference 101) offer the possibility to insert selected biologically effective ions (i.e. Ca^{2+} , F^- , Na^+ , P^+ , etc.).^{102, 103} This technique allows the fine control of the concentration and depth distribution of the implanted elements. However, the potential creation of superficial stresses (removable with a post-annealing)¹⁰¹ and/or modification of preexisting surface nanometric features²¹ must be carefully considered when using this highly-energetic process. Annealing and/or thermal oxidation have been explored on titanium-based metals to enhance their bioactivity by changing the crystalline structure of the nanometric native oxide layer. Previous studies have in fact compared the different crystalline forms of titanium dioxide, and assessed that rutile enhances cellular response.^{104, 105} However, the anatase crystalline phase with a superimposed nanometric topography, generated by DC reactive magnetron sputtering, yielded the best biological results, compared to amorphous TiO_2 and rutile.⁴⁵

3. *In vitro* protein and cellular studies

3.1 Biological surface science

The scientific rationale behind the application of nanotechnology to biomedical surface science correlates with the capacity of cells to sense and recognize specially designed substrate features. Living cells are composed of various structural and molecular elements, synergistically interconnected in a hierarchical system capable of relating events occurring at different levels (i.e. from the molecular to the micrometric scale) and reacting

accordingly. In fact, initial cell-substrate interactions take place at the molecular level, but their effects propagate efficiently throughout the entire structure of the cell, ultimately generating a concerted multiscale cellular response.¹⁰⁶ For this reason, the capacity of initiating and controlling cellular reactions at the nanometric level through nanoscale physicochemical cueing translates into the ability to affect and direct the global behavior of the cell.

When a surface is in contact with the biological environment, the first event which takes place prior to protein adsorption and cell colonization is water adsorption (time scale of order of nanoseconds).¹⁰⁷ The properties of the surface water shell dictate the following events, i.e. the adsorption of plasma and extracellular matrix proteins, determining their orientation, coverage and potential denaturation.^{107–109} Ultimately, the resulting protein adlayer will act as a framework on which cells can adhere, spread, migrate and proliferate. However, it is still not clear whether the observed cellular effects are solely mediated by the protein adlayer or whether the physical surface can also provide cues.

3.2 In vitro protein-substrate interactions

In vitro studies have revealed that the adsorbed protein layer is sensitive to specific physicochemical properties of the implant surface. Nanometric features are more effective in dictating protein adsorption and ultimately determining the biochemical characteristics of the adlayer.¹¹⁰ Surface structuring at the nanoscale may result in changes in surface area/energy, distribution of functional groups, hydrophilicity and oxide composition/thickness, which have been shown to be critical factors to control protein adsorption.^{111–114} In particular, surface topography is also known to affect protein orientation and denaturation, which play a fundamental role in determining the outcome of the subsequent cell colonization.^{115–117} It is believed that only topographical features with dimensions similar to those of surface-bound proteins (~10 nm) can significantly affect their morphology and activity.¹¹⁰ There have been several studies to determine how proteins respond to nanometric features of diverse sizes and morphologies.^{115, 118–122} These have concluded that the adsorption of proteins such as fibrinogen, albumin, and fibronectin (FN) is generally enhanced by nanorough surfaces, although the effect is not always significant.¹²³ In the case of fibronectin, the vertical dimension of nanometric surface features seems to be critical in determining its adsorption profile. This protein is differentially responsive to the depth of nanometric cavities and to the height of spherical ordered nanostructures, ultimately determining the size of the focal adhesions (FAs) (see section 3.3.3).^{124, 125}

3.3 In vitro cell-substrate interactions

3.3.1 Considerations about cells for in vitro studies—Various *in vitro* cell models have been used to better understand the effects that nanostructured surfaces exert on cellular reactions. The choice of a specific cell type reflects the biological property under investigation and ultimately, the application envisaged for a material. For instance, osteoblasts (i.e. bone-forming cells) are used to evaluate the capacity of materials to promote bone formation for orthopedic and dental implants. Similarly, fibroblasts (i.e. soft connective tissue-forming cells) have been used to evaluate the ability to promote or limit the growth of fibrous tissues. In addition, endothelial (i.e. vascular lining cells) and smooth muscle cells allow to assess cardiovascular repair potential. Stem cells (i.e. pluripotent undifferentiated cells¹²⁶) are generally used to determine whether a given nanotopography can induce differentiation along selected pathways.

Cells used for *in vitro* experiments generally derive from transformed cell lines. Although these cells in large part reflect the activity of the cell type from which they derive, they may not necessarily yield similar bioactivity outcomes. On the other hand, primary cultures with

cells isolated from tissues are more difficult to grow, generally loose phenotypic specificity as they are passaged, and exhibit greater biological and differentiation variability. The latter, however, reflects more closely the *in vivo* healing situation where cells at various stages of differentiation are found.

3.3.2 Cellular response to nanostructured metals—The short-term (i.e. cell adhesion and proliferation) and long-term (i.e. bone matrix secretion and mineralization) functions of osteoblasts can be stimulated by anodized titanium surfaces.^{40, 62, 127} Such surfaces also sustained the *in vitro* growth and osteogenic differentiation of human skeletal mesenchymal stem cells (hMSC),^{47, 128} and also induced cytoskeletal reorganization and increased the frequency and size of FAs (Figure 4).⁴⁷ Interestingly, the capacity of anodization to precisely control the nanotopographical features permitted to correlate cell function with diameter and spacing of nanotubular structures.^{44, 129} In particular, nanotubes with 15 nm diameter significantly enhance the adhesion, spreading and differentiation of mesenchymal stem cells as compared to nanotubes with 70–100 nm.^{44, 130} In addition, a spacing of 15 nm provided an effective length scale for accelerated integrin clustering and FAs formation (see section 3.3.3).⁴⁴ Other cell types such as chondrocytes (cartilage-forming cells) and keratinocytes (epidermal cells) were shown to better respond to anodized surfaces.^{131, 132} On the other hand, bacterial growth is significantly limited on nanorough titanium surfaces.¹³³ These results suggest that anodization can be an effective tool for cartilage applications and can be exploited to improve current bone-anchored transcutaneous implants, as well as to endow implantable metals with antibacterial properties.

Oxidative nanopatterning confers titanium-based metals the exciting capacity to selectively influence cellular behavior, by favoring the growth of osteoblastic cells while limiting that of fibroblastic ones.^{21, 52} Very interestingly, adhesion of the two cell types after 1 and 4 hours of treatment was similar on smooth and nanoporous surfaces, suggesting that the effects of the nanotexture do not significantly relate to cell adhesion but rather to cell growth and differentiation. Selective adhesion also takes place on carbon nanofiber compacts (surface roughness of about 0.2 nm as determined by SEM stereoimaging), which have been demonstrated to enhance the osteoblastic activity and limit that of smooth muscle cells, fibroblasts and chondrocytes.¹³⁴ The physicochemical cueing resulting from oxidative nanopatterning also impacts on gene and protein expression, leading to enhanced osteogenic cell adhesion, spreading and proliferation.^{50–52, 135, 136} The generated nanoporous surface induces an early upregulation of bone sialoprotein (BSP) and osteopontin (OPN) expression and promotes extracellular fibronectin assembly.^{50, 135, 136} Early enhanced secretion of proteins with cell adhesion capacity and their assembly onto the surface with which cells interact will affect cell dynamics at an early crucial stage when key parameters of tissue healing are decided.^{137–140} In addition, osteogenic cells grown on nanostructured titanium surfaces created by either H₂SO₄/H₂O₂ or NH₄OH/H₂O₂ etching exhibit a significant upregulation of genes associated with cell adhesion and migration,⁵⁰ such as integrin alpha-5 and hyaluronan, as well as an enhanced extracellular fibronectin assembly (Figure 5).^{50, 136} Similarly to anodization, oxidative nanopatterning can harness the power of stem cells by promoting their growth.⁵⁰ Importantly, a differential gene expression profile has been detected for hMSCs and osteogenic cells grown on nanopatterned titanium surfaces.^{50, 141} Also, expression of key bone markers was upregulated, including that of alkaline phosphatase and Runx2.¹⁴¹ These results indicate that anodization and oxidative nanopatterning are likely to have a key impact on regenerative medicine, as a strategy to achieve predictable tissue healing around the implant.

Nanostructured bioactive coatings generated by sol-gel and chemical vapor deposition enhance osteoblastic cell adhesion and proliferation.^{61, 63, 142} Surfaces modified by covalently-bonded bioeffector molecules have the capacity to positively influence *in vitro*

osteogenic functions, mainly cell attachment, adhesion, proliferation and gene expression.^{64, 143–147}

Concomitant with these chemical methods, physical approaches confer an enhanced biocompatibility to implantable materials by creating on their surfaces coatings presenting nanoscale features. For example, PVD was applied to create modify titanium surfaces with different nanotopographical coatings on titanium surfaces for *in vitro* studies aimed at evaluating the effects of surface chemistry and topography on the cellular and/or tissue response.^{98, 148}

The results of *in vitro* studies are summarized in Table 1.

3.3.3 Potential mechanisms for cell adhesion and proliferation at the nanoscale—Cells adhere to and interact with their extracellular environment through focal adhesions (FAs) (Figure 6). These are dynamic clusters - 15–30 nm in dimension¹⁴⁹ - of transmembrane proteins called integrins that anchor cells and transmit mechanical and regulatory signals.¹⁵⁰ Integrins interact intracellularly with cytoskeletal elements and extracellularly with matrix proteins containing an RGD (arginine-glycine-aspartate) motif. The number of attached cells correlates with the average RGD surface density, and higher RGD surface densities lead to an increase in FA formation.¹⁵¹

Threshold dimensions of nanostructures relating to the assembly of FAs have been established (reviewed in reference 152). The observed cellular effects on nanostructured substrates may in part relate to the propensity of the cell membrane to stretch towards points of stability and in so doing trigger nanoscale cytoskeletal reorganization. The sum total of these nanoscale deformations at various points along the cell membrane will trigger typical signaling pathways that regulate cell behavior. The mechanisms by which integrins signal into the cell interior are complex and they involve several different pathways, such as the cytoplasmic protein tyrosine kinase, called focal adhesion kinase (FAK).¹⁵⁰ Nanoscale pits upregulate FAK signaling in osteoblastic cells.¹⁵² Similarly, nanoporous surfaces are believed to significantly affect the activation of GTPase Rho at FAs,^{152, 153} an enzyme known to play a part in the maturation of the adhesive complex by promoting recruitment of actin filaments and integrins to the contact site.^{150, 154}

During interaction with bi-dimensional substrates the cell changes its shape and dynamically develops extensions called *lamellipodia* (membrane veils rich in actin) and *filopodia* (finger-like protrusions) that act as sensing and traction elements to allow cells to crawl and perceive topographical cues.^{151, 153} Nanotopography promotes the formation of these membrane specializations (Figure 7). Changes in cell shape are as influential as signaling molecules in the differentiation process.^{155–159} Indeed, it has been shown that cell shape regulates the switch in commitment of stem cells towards an osteoblastic phenotype by modulating endogenous RhoA activity.¹⁶⁰

4. *In vivo* and clinical applications

4.1 *In vivo* response of nanostructured implants

Significant efforts have been made so far to integrate surface modification approaches with implant manufacturing in order to ensure superior clinical performance of current prostheses.^{161–163} In this context, while the mechanical performance of a device is mainly defined by its macroscopic shape, tissue integration and long-term stability result from the interactions between the implant surface and the surrounding biological environment at the cellular and molecular level.^{164–167} Based on the *in vitro* results, nanostructured metals undoubtedly have the potential to yield a faster and more stable integration of biomedical

implants.^{34, 56} To validate this hypothesis, several nanoscale surface modification approaches are currently under investigation to determine their effects *in vivo* and their potential applicability to commercial implants.^{34, 168–170} Here, we present an overview of methods that have been successfully adapted to nanomodify metallic implants and the resulting animal and clinical outcomes. The results of the *in vivo* studies are summarized in Table 2.

A recent study has exploited controlled anodization to endow the surface of titanium implants with self-organized and highly ordered TiO₂ nanotubes (30 nm diameter). Implanted in the frontal skull of pigs, these nanotubular surfaces stimulated collagen type-I expression by osteoblasts.¹⁶⁸ However, no significant difference in the amount of peri-implant bone formation was observed, suggesting that these nanotubular surfaces ultimately have no impact on osseointegration. These results differ from previous *in vitro*⁴⁰ and *in vivo*¹⁷¹ studies with similarly treated surfaces but different nanotube diameters (i. e. 70–80 nm and 250–800 nm diameter nanotubes, respectively). These concluded that anodized titanium does not impact on the total collagen content but enhances peri-implant bone formation.^{40, 171} Such differences may result from various factors related to the material properties, such as the different surface topography (i. e. nanotube diameter), degree of crystallinity of the titanium dioxide overlayer, or to other factors unique to *in vivo* studies, such as surgical considerations (see discussion). Consistent with the results previously described (see section 2.1), anodic oxidation also allows deposition of nano-hydroxyapatite (nano-HA) coatings (Figure 8).¹⁷² Histomorphometric evaluation showed that nano-HA modified implants have significantly greater bone-to-implant contact (BIC) values compared to acid-treated implants. The enhancement of BIC by hydroxyapatite nanocrystals was confirmed in a different study in which it was also shown that early bone formation depends on the size and diameter of the nanosized hydroxyapatite crystals.¹⁷³ The same research group exploited a sol-gel technique to deposit nanotopographic titanium dioxide coatings that also enhanced BIC.¹⁷⁴

Micro-arc oxidation (MAO) has been used to dope the implant surface with magnesium.^{175, 176} Although this technique does not result in topographical modifications at the nanoscale, the presence of Mg ions yielded a more rapid and stronger bone-integration, as well as higher removal torques than commercial microrough highly-crystalline and phosphate-enriched titanium implants.^{177, 178}

Oxidative nanopatterning with H₂SO₄/H₂O₂ creates nanotopography on bulk metals,⁵⁰ as well as on titanium screw-shaped implants (Figure 9). Oxidative treatment of commercial screw-shaped titanium implants placed in dog mandibles resulted in enhanced BIC values and osteogenesis and this even though no nanotexture was obtained.¹⁷⁹ The authors suggested as an explanation that the passivation of the implant generated chemical conditions affecting the dynamics of the deoxidation/reoxidation processes. This study, however, shows that existing commercial implants can be significantly improved using a simple treatment that does not require elaborate changes in manufacturing.

The combination of TiO₂ blasting and hydrofluoric acid treatment has been used to create a commercial endosseous titanium implant with microrough surfaces with superimposed uncharacterized features ranging between 50 and 200 nm.¹⁸⁰ These implants stimulated osteoblastic gene expression, as well as they enhanced bone formation, osseointegration and bone-implant fixation.^{181–183} While some inflammatory response was reported,¹⁸⁴ the overall success rate was satisfactory, with the majority of implants yielding good osseointegration and stability after one year post-surgery. Noteworthy, high success rate was also reported under more challenging clinical conditions such as early loading.¹⁸⁵

Dual-acid-etching (DAE) followed by discrete-crystalline-deposition (DCD) has been used to create calcium phosphate (CaP) nanocrystals (20–100 nm in size) coated titanium implants.¹⁸⁶ These implants were clinically evaluated in a randomized controlled trial (RCT), which concluded that BIC in the posterior maxilla increased by about 12%.¹⁸⁷ In addition, bone formation around CaP nanocrystal-coated implants was enhanced.¹⁸⁸ The observed capacity to accelerate bone formation is expected to result in a faster healing response which ultimately may permit earlier loadings. A recent study in dogs comparing bone formation around DAE and CaP nanocrystals-coated implants showed that there was no statistical difference in BIC in the early healing phase (1–4 weeks).¹⁸⁹ However, test sites containing implants coated with nanoparticles exhibited less bone resorption.

Another commercial implant, which was however discontinued in 2009, was produced by ion-beam-assisted-deposition (IBAD) of thin (20–500 nm) bioactive Ca- and P-based ceramic layers.¹⁹⁰ With a 300–500 nm thick coating, the BIC value was not significantly affected, but the implants presented higher torque-to-interface fracture levels compared to controls, indicating an improved fixation.¹⁹¹ However, the biomechanical properties degraded when the bioactive layer thickness was reduced to 20–50 nm and there was no improvement of BIC values and bone fixation at early implantation times (2 and 4 weeks).¹⁹⁰ This outcome was partially attributed to the reduced thickness and to the amorphous microstructure of the deposited material, which may have caused its rapid dissolution.

Nanostructured surfaces presenting uncharacterized features of approximately 100 nm have been obtained on microrough titanium implants by hydrothermal alkaline treatment.¹⁹² This method also caused incorporation of calcium as CaTiO₃ in the protective surface oxide layer. Titanium and Ti6Al4V implants with thus treated surfaces significantly increased BIC and removal torque forces in rabbit tibiae.^{171, 193, 194} In addition, this treatment improved bone formation around implants placed in trabecular bone.¹⁹²

Bioceramic grit-blasting and acid etching (BGB/AA) technologies have been integrated to produce submicrometric topography on titanium implants.¹⁹⁵ Histological evaluation after 2 months post-surgery showed significantly higher BIC and osteocyte density around modified implants when compared to dual acid-etched implants.¹⁹⁶

4.2. Antibacterial and cardiovascular implants

Infection at the implantation site represents an important clinical complication. In the United States, approximately 4.3% of the orthopedic implants developed bacterial infection. In the case of cardiovascular devices, the annual infection rate is even higher (7.4%).¹⁹⁷ Coatings deposited by chemical (e.g. wet-chemistry approaches) and physical (e.g. PVD) processes (reviewed in reference 198) have been successfully applied to control bacterial colonization of implanted prostheses. For example, biodegradable gentamicin-loaded poly(D, L-lactide) (PDLLA) coatings prevented implant-related *osteomyelitis* (i.e. infection of bone) around titanium wires.¹⁹⁹

In this context, while titanium-based metals are the most common material for dental and orthopedic applications, their use in cardiology is still limited.²⁰⁰ Stainless steels, Ni-Ti (Nitinol) and Co-Cr alloys, as well as Tantalum, are in fact more often encountered in cardiovascular applications.²⁰⁰ Degradable magnesium- and iron-based alloys are also now being considered in cases when the implant is only temporarily needed, such as for stents.^{12, 13}

Stents capable of slowly releasing drugs (i.e. immunosuppressant, chemotherapeutic, etc), for example to block cell proliferation or prevent the formation of scar tissue at the site of

coronary intervention, are already approved and commercially-available (drug-eluting stents, DES).²⁰¹ However, the polymeric coating used in DES may cause short-term inflammation as well as long-term thrombotic events.²⁰² Nanostructured may represent an alternative such polymer-based DES. In this context, nanoporous surfaces generated on coronary stents by various techniques yielded promising results in terms of drug-eluting capacities (reviewed in reference 202) (Figure 10). For example, a nanostructured stent was developed by coating a 316L stainless steel with electrochemically-generated nanoporous Al_2O_3 .²⁰³ However, the clinical performance of this system was not satisfactory.²⁰⁴ In addition, nanoporous metallic coatings were created through sputtering techniques, which allow deposition of a variety of materials, including cobalt chromium, nitinol, platinum, and stainless steel.²⁰² Oxidative nanopatterning has also been efficiently applied to sputtered titanium to create a network of nanopores.¹²² Therefore, deposition of thin films of titanium on substrates that are not readily amenable to chemical oxidation could be advantageously exploited to create nanotextures offering antiadhesive properties,⁵⁰ thus providing a valuable strategy to improve the integration of stents made of different metals.

4.3. Micromotion and nanotopography

The primary stability of an implant results from the biomechanical interlocking of the implant and the surrounding bone. In addition to implant-related factors (material, design, topography and surface chemistry), surgical procedure and patient variable (bone quantity and quality, health condition),²⁰⁵ the local mechanical environment has been recognized as a critical factor for bone repair and regeneration around an implant.²⁰⁶ However, the exact biomechanical parameters involved in the initial healing process at the tissue-implant interface remain elusive. Several studies have demonstrated that micromotion during implant loading, can arise from various factors including: (1) the physical properties of the bone tissue (e.g. elastic moduli of adjacent bone and strength of attachment between implant and tissue), (2) the mechanical interaction between the implant and tissue, (3) the initial fit of the implant in the site, (4) the geometry of the implant,^{207–209} and of course (5) the characteristics of the loading.^{206, 208} Depending on the magnitude and direction of resulting micromotion at the interface, tissue differentiation around immediately loaded implants can respond differently; excessive micromotion may be a compromising factor that leads to interfacial fibrous tissue formation and prevention of osseointegration, while at the lower range, there is evidence that there may be positive effects on bone repair and regeneration.^{210–215} Overall, the exact dynamics of the interfacial events that lead to osseointegration when healing occurs under loading and the limits of relative motion as well as the specific aspects of micromotion that actually cause problems at the bone-implant interface still have to be established.

Implant micromotion *per se* may not be the decisive factor in interfacial healing; instead, there is now evidence that the motion-associated interfacial principal strain magnitude – along with its spatial distribution, duty cycle, waveform, etc. – is the factor that most influences osseointegration.^{216, 217} This is a significant finding because of the current debates surrounding ‘immediate’ vs. ‘delayed’ loading of implants²¹⁸, recall that implant loading inevitably creates strain fields at the interface, whether the loading starts immediately after implantation, or later, after the bone has had a chance to heal to some degree. Therefore, as one contemplates strategies for the design and use of dental and orthopedic implants, it is not the timing of loading that is the primary issue, but rather the propensity of the implant to undergo micromotion and thereby create strain in the local interfacial. Although previous studies have tried to derive a single value of limiting strain and stress levels beyond which bone regeneration is compromised,^{206, 219} it may not be possible to define such a value, even the exact type of stress or strain that is key is also unclear. It is well to remember that continuum-based models of mechanotransduction will

always have limitations when it comes to unraveling detailed molecular-level mechanisms at the cell level, as elegantly discussed by Humphrey.²²⁰ However, based on ongoing studies on a mouse tibial model, it has been estimated that ~30% principal strain (tensile or compressive) represented an upper limit to allowable principal strain magnitude during interfacial bone healing.^{217, 221} There are also orthopedic studies from the fracture healing and dynamization literature that also support this view, although again, exact thresholds remain elusive.^{222–224} Since an appreciable number of implant losses have been considered to possibly be the result of excessive strains and stresses at the bone-implant interface, a better knowledge of the interfacial stress-strain conditions that are tolerable as opposed to deleterious interfacial stress-strain conditions is essential, and would lead to understanding the mechanically-controlled tissue differentiation process, that is the “mechanobiology” at the interface.

On question that has received relatively less attention is whether surface characteristics have an impact on the interfacial mechanobiology. This aspect should be considered at multiscale levels, including the macro, micro and even the nano levels. Macro- and micron-scale surface modifications have been exploited for some years now to create biomechanical environments that can significantly modulate tissue healing around bone implants. For example, there is a long history of exploring “optimal” porous implant designs for bony ingrowth as well as attempts to understand the role of surface roughness at the micron-scale in terms of implant accommodation in bone.²²⁵ The more recent recognition that interactions between biomaterials and host tissues also take place on the nanoscale has focused attention of researchers and industry to a new generation of implantable metals with nanoengineered surfaces. This has in part been driven by intriguing work suggesting that an implant’s surface texture or roughness may alter cell shape and cytoskeleton, which in turn may influence gene expression.^{226–229}

As a consequence, there is already a growing body of literature exploring the role of nano-level mechanics on cell-substrate interactions as well as cell-cell interactions. For example, recent reviews indicate that substrate elasticity as well as mechanical deformations (strain) of cells play a role in determining stem cell fate decisions.^{106, 230} This latter perspective naturally leads to the idea that the nano-level mechanical properties of an implanted biomaterial could be influential in determining cell fate decisions adjacent to a biomaterial. As stated by Discher *et al.* “... not only physical contributions to differentiation but also that carefully made materials can help prime the expansion of specific progenitors.”²³⁰ Finally, this last perspective also leads to the notion that finite element analyses and similar mechanical modeling methods could help take account of the truly multiscale character of the biomechanics problem at bone-implant interfaces; such models could be helpful to fully scope out the possible interplay of cell deformations in relation to implant surface topography.¹⁰⁶

Given this body of work indicating that physical factors at the tissue and cell levels, including roughness, mechanical properties, as well as related stresses and strains, are ultimately transferred to single cells in some manner, it is clear that local strain fields at the macroscale will then affect cellular events at the micron- and nano- (molecular) scale.²³¹ Yet from both mechanistic and implant-design viewpoints, there is little to bridge the gap between implant design at macro- and micron-levels vs. biological mechanisms at the nanoscale. Hence it can be hypothesized that local deformation in healing tissue (e.g. by micromotion) generates changes in the biomechanical relationships between tissue and implant at multiple size scales (macro, micro and nano) and therefore influences local cellular (e.g., cell morphology) and molecular (e.g. gene expression) events. In turn, surface features on the implant may also participate in creating strain fields that regulate cell fate decisions at the bone-implant interface. In fact, a series of nanoscale cell distortions might

cumulatively create substantial strain to trigger changes in cytoskeletal organization and activation of different signaling pathways in diverse cell types.²³² Since nanoscale topography and chemistry have been demonstrated *in vitro* to influence the mechanical properties of cells,^{233, 234} it seems likely that this could be advantageously exploited *in vivo* to allow the design of an implant that might counterbalance negative effects of 'excessive' micromotion. For example, nanoscale cell distortions induced by micromotion could be exploited to selectively trigger cell fate decisions (e.g. osteogenic vs. fibroblastic pathway)²³² thereby minimizing fibrous tissue formation and promoting bone formation around loaded implants.

4.4. Considerations on the biocompatibility and cytotoxicity of implantable metals

The cytotoxicity of implanted metals is a major concern in the biomedical field and still object of debate.²³⁵ Ions and/or debris released from an implanted material can in fact provoke adverse immune and inflammatory reactions that can ultimately defeat the purpose of the biomedical device. It is commonly accepted that titanium and its alloys are biocompatible. However, some titanium alloys show a certain degree of cytotoxicity due to the alloying elements.²³⁶ For example, Al and V were associated to adverse reactions in tissues and neurological disorders.²³⁷ In addition, high levels of Cr, Ni and Co raise the risk of carcinogenicity.²³⁶ While these effects are known, it is important to point out that the links between a precise concentration of metallic ions and adverse phenomena such as DNA damage, DNA mutation, carcinogenicity and metal sensitivity have not yet been unequivocally established,²³⁵ and further studies are thus required.

5. Outlook and perspectives

In this paper, we have presented an overview of current nanotechnology-based strategies (chemical, biochemical and physical) which have been used to improve the *in vitro* biological response to implantable metals, and in some cases have also shown enhanced *in vivo* experimental outcomes and found commercial applications. In general, while it can be concluded that nanoscale surface modification introduces novel bioactive capacity into the arena of metallic biomaterials, many of the modification approaches described in this review still remain to be tested *in vivo* and few have actually been exported to the medical device industry. There are in fact comparatively very few structured animal and clinical studies that investigated the short and especially long-term effects of such surfaces, evaluating whether the enhanced biological activities demonstrated *in vitro* actually translate to the complex *in vivo* environment. However, even though it is well known in biology that *in vitro* conditions do not fully reflect the *in vivo* reality, such studies are nonetheless essential to screen potentially successful treatments for subsequent animal validation.

From the material point of view, it should be noted that most *in vitro* studies aimed at probing in detail how cells respond to a given nanostructured substrate have been carried out on non-metallic materials, generally characterized by nanoscale protrusions created by techniques such as colloidal and electron beam lithography, and polymer phase separation.¹⁵² There are comparatively few studies that have investigated pits which fit the commonly-accepted nanometric definition (i.e. with dimension smaller than 100 nm),^{125, 238} on implantable metals. While studies carried out with laboratory materials are undoubtedly important to understand the dynamics involved in cell adhesion to substrates, the resulting cell behaviors may ultimately differ from those on medically-relevant metals which exhibit different surface physicochemical parameters. In this context, anodic oxidation and chemical nanopatterning are efficient tools to create networks of nanometric pits (less than 100 nm in diameter) on implantable metals to probe cell adhesion.^{47, 50, 135, 239}

Since in most cases cell cultures on nanopatterned materials and, as a matter of fact even on microstructured surfaces, are carried out in the presence of proteins, one important question that remains to be addressed is whether there is a direct contribution by surface modification. It is now well established that proteins rapidly form an adsorbed layer on biomaterial surfaces, and that these proteins affect diverse cellular events at the interface, including cell morphology, growth, and function.^{107–109} It is therefore not clear whether the resulting *in vitro* effects are due to direct signaling by the nanometric surface cues, to the protein adlayer or the synergistic combination of these two factors. It can, in any case, ultimately be argued that the cellular outcomes result from the nanometric surface features since these significantly affect protein morphology and activity.^{107, 118–121} Irrespectively, there is another important consideration that needs to be made when translating *in vivo* protein adsorption profiles to the body. The identity of the first proteins to adsorb onto the surface and subsequent protein-protein interactions may not be precisely predicted due to the complexity, variability and dynamism of the biological environment. In multi-component solutions such as bodily fluids, competitive adsorption and protein displacement concur in determining the final adsorption profile.^{240, 241} For instance, with time high-molecular weight proteins can displace smaller proteins already adsorbed onto a substrate. Therefore, one would expect that overall a multi-component mixture will interact in a semi-stochastic manner with the substrate, inducing a certain degree of variability in the overall adsorption profile. In addition, in the body at sites where there is bleeding, a fibrinogen coagulate will form rapidly and may partially or completely mask the surface thereby complicating further the formation of an adlayer on an implant.²⁴² One must be therefore extremely careful in extrapolating data obtained *in vitro* to the reality of the biological environment where the behavior of a protein will not only depend on its physicochemical properties and those of the biomaterial surface, but will also reflect modulation of their behavior by the various tissue and body fluid components it is exposed to.²⁴³ Predictable adsorption may ultimately require more selective approaches such as nanopatterning of surfaces with molecular arrays that will determine interactions of the substrate with selected proteins.

There are additional variables determining the final outcome of an implant, such as the implantation site and the overall health condition of the patient. Longitudinal and descriptive human studies, as well as histological data, have indicated that early osseointegration and clinical success rates for endosseous implants vary according to the anatomic location.²⁴⁴ A high success rate has been reported for implants placed in regions with dense bone such as the mandible,^{245, 246} whereas placement in the maxilla can yield insufficient osseointegration, partially related to poorer bone quality at this site.²⁴⁷ Difficulty in achieving short- and long- term integration and stability of dental implants has also been observed in selected patient populations, such as smokers and diabetics.^{248, 249} Due to the complexity of the clinical scenario, new nanotextured implants may not necessarily yield the expected outcome when used clinically and longer evaluation periods may be required.²⁵⁰ As stated by Albrektsson, “novel implant systems or moderate/major changes of existing implants need to be clinically tested preferably before clinical introduction, not afterwards”.²⁵¹ While clearly caution is needed, the animal and clinical studies carried out so far suggest that nanoscale surface modification of metallic implants has a promising clinical future.

A particular problem is that implant studies seldom provide a detailed characterization of surfaces at the nanoscale.²⁵² As stated by Wennerberg, measurements and evaluation techniques need to be standardized.²⁵³ For example, roughness measurements as well as protocols to determine surface wettability should be more uniform in order to provide the consistent data that is required for reliable comparison of different implants.²⁵⁴ In this context, contact angle measurements have been extensively used to determine the wettability and free energy of nanostructured substrates^{53, 255} and characterize biomedical implants,

since these parameters affect the interactions between a material and the surrounding biological environment.^{256–259} Experimental conditions (e.g. drop size and symmetry, vibration of the substrate), as well as the physicochemical properties of the substrate (e.g. surface roughness, porosity, chemical homogeneity and presence of adsorbates) and the presence of surface nanostructures, are all factors that influence the overall wettability of a material, and should therefore be carefully taken into consideration during data analysis.^{258–260} In addition, some prosthetic devices commercially-advertised as nanostructured are characterized by inhomogeneous nanofeatures which lack statistical distribution and/or precise reproducibility. In this context, even smooth implant surfaces could be considered nanostructured since they could actually exhibit a nanoscale roughness or some kind of nanometric features. These are, most of the times, the direct consequence of the fact that materials are not naturally atomically flat and characterized by defects. Therefore, the need of a distinction between engineered nanostructures and the native topography of materials at the nanoscale, is legitimate. All these factors may introduce additional variables which, in addition to biological variability, may ultimately confound assessment of the precise role of nanometric surfaces in controlling biological events in living systems. It will be therefore important to optimize current and future nanotechnology approaches to generate surfaces with controlled and uniform surface nanofeatures in order to achieve a better understanding of material/host tissue interactions.

While the majority of studies with nanostructured metals has focused on dental and orthopedic implants, the benefits of nanoscale surface modifications for cardiovascular applications have just started to be assessed.^{130, 261, 262} A problem associated with current drug-eluting stents (DESs) is the potential cracking/detachment of the polymeric coating during balloon expansion.²⁶³ In addition, delayed healing and hypersensitivity reaction to DES polymeric components have been reported.²⁶⁴ Direct nanoscale modification of surfaces without deposition of polymeric overlayers could represent an effective strategy to limit adverse reactions to metallic stents. As an example, porous surfaces capable of selectively influencing cell growth (e.g. chemically-generated nanoporous surfaces⁵⁰) could improve the performance of existing stents by hampering undesired cell adhesion and proliferation, and also improve tissue healing by synergistic drug release (see below).

Nanostructured materials are extensively used as biologically active molecule-delivering systems.²⁶⁵ Compared to surfaces with nano-protrusions, nanoporous surfaces are very attractive because they 1) offer an increased surface area which translates into a greater surface reactivity, and 2) provide nanoconfined volumes which are expected to modulate dissolution rates. The fact that higher area-to-volume ratios translate into a higher amount of surface reactive sites is well known in physical chemistry and biology, since it determines, for instance, catalytic reaction rates and extracellular substance exchange, among others. On the other hand, nanoconfined volumes influence physicochemical events such as fluid mobility.²⁶⁶ Taken together, these factors indicate that nanoporous surfaces have the potential for enhanced loading ability and the capacity to provide controlled dissolution rates over time.

The greater surface area provided by nanopores can be synergistically exploited to link more efficiently bioactive molecules to metals, with or without the use of intermediate linking agents such as oxylanes and phosphonates.²⁶⁷ To this end, it will be interesting to compare the outcome of smooth and nanostructured surfaces with covalently-linked synthetic peptides known to promote cell adhesion and osteoblastic differentiation such as (1) the noncollagen-derived RGD,²⁶⁸ and (2) the collagen-derived P-15,²⁶⁹ DGEA²⁷⁰ and GFOGER²⁷¹ collagen-related cell binding peptides. However, the best linking agents for bioeffector molecules are likely to be more complex molecules, capable of forming multiple stable covalent links (e.g. multidentate molecules) onto metallic surfaces while creating

lateral self-assembly. This is also expected to result in the capacity of generating various molecular patterns, with tunable properties (such as the pattern geometry, nature and orientation of active groups), even on surfaces which are not flat at the atomic scale (such as those of commercially-available implantable metals). In this way, it will be possible to extend the panoply of tailored nanostructures by designing organic patterns otherwise non-easily achievable by direct physicochemical methods. In addition, we can envisage molecular arrays capable of responding to specific variations of the surrounding biological environment, such as local changes in the pH or temperature at the material-host tissue interface, paving the way for novel biomaterials.

We believe that future generations of implants should aim at “intelligent” surfaces, capable not only of providing cues to the surrounding tissues but also of receiving and responding to events occurring in them. Here, we anticipate that synergistic approaches will be critical to achieve the challenging goal of endowing implantable materials with the capacity to “dialogue” with the surrounding environment. For example, nanoporous surfaces could be further functionalized with molecular arrays or multi-layered coatings that would in a first step provide signaling to cells and that could in-time respond to local changes in the environment, exposing underlying nanopatterns capable of physical cueing or drug release. It could be conceived that molecular reorganization could also be initiated by external stimuli, such as infrared light, ultrasounds and magnetic fields, for instance. Similarly, functionalized nanoparticles could be embedded in molecular arrays sensitive to local chemical variations.

There is also limited information on whether the release of toxic metallic ions can be limited by nanoengineered surfaces. For example, chemical treatments that, besides generating nanoscale cues, yield an increase in the protective oxide layer could be a possible solution to better ‘insulate’ metals from the surrounding biological environment. In addition, it will be interesting to determine whether surface nanostructures can by themselves altering the dynamics of dissolution and thus reduce cytotoxic effects. This could be the case for nanoconfined volumes such as nanotubes and nanoporosity, coatings with HA nanocrystals, and functionalization with molecular arrays.

A major challenge in regenerative medicine is the capacity to attract local stem cells and to induce their proliferation and differentiation. Physicochemical surface characteristics may offer an interesting strategy to achieve such a capacity around implantable metals and other biomaterials. In addition, biochemical functionalization could be an efficient complement to physical cueing. In this context, *in vitro* studies have already shown that nanoscale surface cueing can stimulate stem cell growth and guide their differentiation without exposure to molecular signals.^{50, 272} Although this aspect is just beginning to be explored and still lacks *in vivo* validation, current results are certainly promising for skeletal applications. An exciting new development in biology is the finding that stem cells can also be obtained from skin.²⁷³ Bone reconstruction around an implant could thus be induced by exploiting the ability of nanostructured surfaces to activate stem cells and to accelerate bone formation while avoiding tissue encapsulation.^{51, 52, 274} This will translate into faster healing and improved stability of dental and orthopedic prostheses, critical aspects especially in aging patients with poor bone support.

In conclusion, nanoscale surface modification approaches are likely to foster profound changes in the ways implants and other biomaterials are designed and manufactured. The application of medical prostheses to improve health and even save lives will continue to grow until tissue engineering reaches a mature level and regeneration of complete tissues can be achieved. Therefore, any procedure that can improve the performance of implantable biomaterials will have a major impact on the quality of life and result in a major economic

benefits to society. Advances in metallurgical and surface-engineering techniques, and nanotechnology promise a new generation of improved prosthetic devices with selective bioactive surfaces, and eventually with “intelligent surfaces” capable of responding to the implantation site environment.

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References

1. Kurtz S, Ong K, Lau E, Mowat F, Halpern M. J Bone Joint Surg Am. 2007; 89:780. [PubMed: 17403800]
2. Hodgson JM, Bottner RK, Klein LW, Walpole HT, Cohen DJ, Cutlip DE, Fenninger RB, Firth BG, Greenberg D, Kalisky I, Meskan T, Powell W, Stone GW, Zito JP, Clark MA. Catheter Cardiovasc Interv. 2004; 62:1. [PubMed: 15103593]
3. Kereiakes DJ, Choo JK, Young JJ, Broderick TM. Rev Cardiovasc Med. 2004; 5:9. [PubMed: 15029110]
4. Wong, JY.; Bronzino, JD. Biomaterials. CRC Press; Boca Raton USA: 2007.
5. Ratner, BD.; Hoffman, AS.; Schoen, FJ.; Lemons, JE. Biomaterials Science: An Introduction to Materials in Medicine. Elsevier Academic Press; San Diego USA: 2004.
6. Brown, SA.; Lemons, JE. Medical Applications of Titanium and Its Alloys: The material and biological issues. West Conshohocken USA: 1996.
7. Mani G, Feldman MD, Patel D, Agrawal CM. Biomaterials. 2007; 28:1689. [PubMed: 17188349]
8. Bhat, SV. Biomaterials. London UK: 2002.
9. Black, J. Orthopedic biomaterials in research, practise. Livingstone; New York: 1998. Corrosion, degradation.
10. Bobyn JD, Stackpool GJ, Hacking SA, Tanzer M, Krygier JJ. J Bone Joint Surg B. 1999; 81:907.
11. Witte F. Acta Biomater. 2010; 6:1680. [PubMed: 20172057]
12. Purnama A, Hermawan H, Couet J, Mantovani D. Acta Biomater. 2010; 6:1800.
13. Hermawan H, Dubé D, Mantovani D. Acta Biomater. 2010; 6:1693. [PubMed: 19815097]
14. Liu H, Webster TJ. Biomaterials. 2007; 28:354. [PubMed: 17049982]
15. Mark, Kvd; Bauer, JPS.; Schmuki, P. Cell Tissue Res. 2010; 339:131. [PubMed: 19898872]
16. Horley GA. Small. 2006; 1:3.
17. Whitesides GM. Nat Biotechnol. 2003; 21:1161. [PubMed: 14520400]
18. Rho JY, Spearing LK, Zioupos P. Med Eng Phys. 1998; 20:92. [PubMed: 9679227]
19. Sniadecki NJ, Desai RA, Ruiz SA, Chen CS. Ann Biomed Eng. 2006; 34:59. [PubMed: 16525764]
20. Stevens MM, George JH. Science. 2005; 310:1135. [PubMed: 16293749]
21. Variola F, Vetrone F, Richert L, Jedrzejowski P, Yi J-H, Zalzal S, Clair S, Sarkissian A, Perepichka DF, Wuest JD, Rosei F, Nanci A. Small. 2009; 5:996. [PubMed: 19360718]
22. Barbucci R, Pasqui D, Wirsén A, Affrossman S, Curtis A, Tetta C. J Mater Sci-Mater Med. 2003; 14:721. [PubMed: 15348415]
23. Liu X, Chu PK, Ding C. Mater Sci Eng R. 2004; 47:49.
24. Guizzardi S, Galli C, Martini D, Belletti S, Tinti A, Raspanti M, Taddei P, Ruggeri A, Scandroglio R. J Periodontol. 2004; 75:273. [PubMed: 15068116]
25. Anselme K, Biggerelle M. Acta Biomater. 2005; 1:211. [PubMed: 16701798]
26. Biggerelle M, Anselme K, Noel B, Ruderman I, Hardouin P, Iost A. Biomaterials. 2002; 23:1563. [PubMed: 11922462]

27. Anselme K, Biggerelle M. *J Mater Sci-Mater Med*. 2006; 17:471. [PubMed: 16688588]
28. Kim M-J, Choi M-U, Kim C-W. *Biomaterials*. 2006; 27:5502. [PubMed: 16857255]
29. Aparicio C, Gil FJ, Planell JA, Engel E. *J Mater Sci: Mater Med*. 2002; 13:1105. [PubMed: 15348651]
30. Park J-W, Jang I-S, Suh J-Y. *J Biomed Mat Res B*. 2008; 84:400.
31. Daugaard H, Elmengaard B, Bechtold JE, Soballe K. *J Biomed Mat Res A*. 2008; 87:434.
32. Piattelli A, Celletti R, Marinho VC, Traini T, Orsini G, Bracchetti G, Perrotti V. *J Long Term Eff Med Implants*. 2006; 16:131. [PubMed: 16700653]
33. Piattelli M, Scarano A, Paolantonio M, Iezzi G, Petrone G, Piattelli A. *J Oral Implantol*. 2002; 28:2. [PubMed: 12498456]
34. Khang D, Lu J, Yao C, Haberstroh KM, Webster TJ. *Biomaterials*. 2008; 29:970. [PubMed: 18096222]
35. Mendonça G, Mendonça DBS, Aragao FJL, Cooper LF. *Biomaterials*. 2008; 29:3822. [PubMed: 18617258]
36. Kim K-H, Ramaswamy N. *Dent Mater J*. 2009; 28:20. [PubMed: 19280965]
37. Ercan B, Webster TJ. *Biomaterials*. 2010; 31:3684. [PubMed: 20149926]
38. Balasundaram G, Yao C, Webster TJ. *J Biomed Mat Res A*. 2008; 84:447.
39. Macak JM, Tsuchiya H, Taveira L, Ghicov A, Schmuki P. *J Biomed Mat Res A*. 2005; 75:928.
40. Yao C, Slamovich EB, Webster TJ. *J Biomed Mat Res A*. 2008; 85:157.
41. Park J, Bauer S, Schlegel KA, Neukam FW, Mark Kvd, Schmuki P. *Small*. 2009; 5:666. [PubMed: 19235196]
42. Oh S-H, Finones RR, Daraio C, Chen L-H, Jin S. *Biomaterials*. 2005; 26:4938. [PubMed: 15769528]
43. Bauer S, Kleber S, Schmuki P. *Electrochem Commun*. 2006; 8:1321.
44. Park J, Bauer S, Mark Kvd, Schmuki P. *Nano Lett*. 2007; 7:1686. [PubMed: 17503870]
45. He J, Zhou W, Zhou X, Zhong X, Zhang X, Wan P, Zhu B, Chen W. *J Mater Sci: Mater Med*. 2008; 19:3465. [PubMed: 18592349]
46. Roy SC, Paulose M, Grimes CA. *Biomaterials*. 2007; 28:4667. [PubMed: 17692372]
47. Sjostrom T, Dalby MJ, Hart A, Tare R, Oreffo ROC, Su B. *Acta Biomater*. 2009; 5:1433. [PubMed: 19208503]
48. Narayanan R, Kim S-Y, Kwon T-Y, Kim K-H. *J Biomed Mater Res Part A*. 2008; 87:1053.
49. Lin C, Han H, Zhang F, Li A. *J Mater Sci- Mater Med*. 2008; 19:2569. [PubMed: 17619960]
50. Vetrone F, Variola F, Oliveira PTd, Zalzal SF, Yi J-H, Sam J, Bombonato-Prado KF, Sarkissian A, Perepichka DF, Wuest JD, Rosei F, Nanci A. *Nano Letters*. 2009; 11:659. [PubMed: 19159323]
51. Variola F, Yi J-H, Richert L, Wuest JD, Rosei F, Nanci A. *Biomaterials*. 2008; 29:1285. [PubMed: 18155762]
52. Richert L, Vetrone F, Yi J-H, Zalzal SF, Wuest JD, Rosei F, Nanci A. *Adv Mat*. 2008; 20:1488.
53. Variola F, Lauria A, Nanci A, Rosei F. *Adv Eng Mat*. 2009; 11:B227.
54. Yoshinari M, Oda Y, Kato T, Okuda K. *Biomaterials*. 2001; 22:2043. [PubMed: 11426884]
55. Cooper LF, Zhou Y, Takebe J, Guo J, Abron A, Holmén A, Ellingsen JE. *Biomaterials*. 2006; 27:926. [PubMed: 16112191]
56. Smith LJ, Swaim JS, Yao C, Haberstroh KM, Nauman EA, Webster TJ. *Int J Nanomedicine*. 2007; 2:493. [PubMed: 18019847]
57. Rani VVD, Manzoor K, Menon D, Selvamurugan N, Nair SV. *Nanotechnology*. 2010; 20:195101.
58. Eisenbarth E, Velten D, Breme J. *Biomol Eng*. 2007; 24:27. [PubMed: 16828342]
59. Popescu S, Demetrescu I, Sarantopoulos C, Gleizes AN, Iordachescu D. *J Mater Sci-Mater Med*. 2007; 18:2075. [PubMed: 17562134]
60. Eisenbarth E, Velten D, Muller M, Thull R, Breme J. *J Biomed Mat Res A*. 2006; 79:166.
61. Amaral M, Dias AG, Gomes PS, Lopes MA, Silva RF, Santos JD, Fernandes MH. *J Biomed Mat Res A*. 2008; 87:91.

62. Das T, Ghosh D, Bhattacharyya TK, Maiti TK. *J Mater Sci- Mater Med*. 2007; 18:493. [PubMed: 17334700]
63. Roy RK, Lee K-R. *J Biomed Mat Res B*. 2007; 83:72.
64. Morra M. *Eur Cell Mater*. 2006; 12:1. [PubMed: 16865661]
65. Beutner R, Michael J, Schwenzer B, Scharnweber D. *J R Soc Interface*. 2010; 7:S93. [PubMed: 19889692]
66. Schliephake H, Scharnweber D. *J Mater Chem*. 2008; 18:2404.
67. Bierbaum S, et al. *J Biomed Mat Res A*. 2003; 67:421.
68. Ku Y, Chung C-P, Jang J-H. *Biomaterials*. 2005; 26:5153. [PubMed: 15792542]
69. Junker R, Dimakis A, Thoneick M, Jansen JA. *Clin Oral Implants Res*. 2009; 20:185. [PubMed: 19663965]
70. O'Toole GC, Salih E, Gallagher C, FitzPatrick D, O'Higgins N, O'Rourke SK. *J Orthop Res*. 2004; 22:641. [PubMed: 15099646]
71. Wikesjö UM, Huang YH, Xiropaidis AV, Sorensen RG, Rohrer MD, Prasad HS, Wozney JM, Hall J. *J Clin Periodontol*. 2008; 35:992. [PubMed: 18976396]
72. Schouten C, Meijer GJ, Beucken JJvd, Spauwen PH, Jansen JA. *Clin Oral Implants Res*. 2009; 20:421. [PubMed: 19298297]
73. Susin C, Qahash M, Polimeni G, Lu PH, Prasad HS, Rohrer MD, Hall J, Wikesjö UM. *J Clin Periodontol*. 2010; 37:574. [PubMed: 20345394]
74. Bussy C, Verhoef R, Haeger A, Morra M, Duval JL, Vigneron P, Bensoussan A, Velzenberger E, Cascardo G, Cassinelli C, Schols H, Knox JP, Nagel MD. *J Biomed Mater Res A*. 2008; 86:597. [PubMed: 18004795]
75. Bumgardner JD, Chesnutt BM, Yuan Y, Yang Y, Appleford M, Oh S, McLaughlin R, Elder SH, Ong JL. *Implant Dent*. 2007; 16:66. [PubMed: 17356373]
76. Greene AH, Bumgardner JD, Yang Y, Moseley J, Haggard WO. *Clin Orthop Relat Res*. 2008; 466:1699. [PubMed: 18443893]
77. Caro A, Humblot V, Méthivier C, Minier M, Salmain M, Pradier CM. *J Phys Chem B*. 2009; 113:2101. [PubMed: 19166331]
78. Cardenas G, Anaya P, Plessing Cv, Rojas C, Sepulveda J. *J Mater Sci-Mater Med*. 2008; 19:2397. [PubMed: 18165888]
79. Martin HJ, Schulz KH, Bumgardner JD, Walters KB. *Langmuir*. 2007; 23:6645–6651. [PubMed: 17488131]
80. Antoci V, Adams CS, Parvizi J, Ducheyne P, Shapiro IM, Hickok NJ. *Clin Orthop Relat Res*. 2007; 461:81. [PubMed: 17549031]
81. Antoci V, King SB, Jose B, Parvizi J, Zeiger AR, Wickstrom E, Freeman TA, Composto RJ, Ducheyne P, Shapiro IM, Hickok NJ, Adams CS. *J Orthop Res*. 2007; 25:858. [PubMed: 17415753]
82. Darouiche RO, Mansouri MD, Zakarevicz D, AlSharif A, Landon GC. *J Bone Joint Surg Am*. 2007; 89:792. [PubMed: 17403802]
83. Chun AL, Moralez JG, Webster TJ, Fenniri H. *Biomaterials*. 2005; 26:7304. [PubMed: 16023193]
84. Chun AL, Moralez JG, Fenniri H, Webster TJ. *Nanotechnology*. 2004; 15:S234.
85. Jansson E, Tengvall P. *Colloids Surf B*. 2004; 35:45.
86. Muller R, Abke J, Schnell E, Scharnweber D, Kujat R, Englert C, Taheri D, Nerlich M, Angele P. *Biomaterials*. 2006; 27:4059. [PubMed: 16580064]
87. Nanci A, et al. *J Biomed Mat Res*. 1998; 40:324.
88. Toworfe GK, Bhattacharyya S, Composto RJ, Adams CS, Shapiro IM, Ducheyne P. *J Tissue Eng Reg Med*. 2009; 3:26.
89. Viornery C, et al. *J Biomed Mat Res*. 2002; 62:149.
90. Falconnet D, Csucs G, Grandin HM, Textor M. *Biomaterials*. 2006; 27:3044. [PubMed: 16458351]
91. Monchaux E, Vermette P. *J Biomed Mater Res A*. 2008; 85:1052. [PubMed: 17937414]
92. Sask KN, Zhitomirsky I, Berry LR, Chan AK, Brash JL. *Acta Biomater*. 2010 in Press.

93. Jeyachandran YL, Mielczarski JA, Mielczarski E, Rai B. *J Colloid Interface Sci.* 2010; 341:136. [PubMed: 19818963]
94. Kim TG, Lee Y, Park TG. *Int J Pharm.* 2010; 384:181. [PubMed: 19799974]
95. Mahapatro A, Johnson DM, Patel DN, Feldman MD, Ayon AA, Agrawal CM. *Curr Top Med Chem.* 2008; 8:281. [PubMed: 18393891]
96. Nakano K, Egashira K, Masuda S, Funakoshi K, Zhao G, Kimura S, Matoba T, Sueishi K, Endo Y, Kawashima Y, Hara K, Tsujimoto H, Tominaga R, Sunagawa K. *JACC Cardiovasc Interv.* 2009; 2:277. [PubMed: 19463437]
97. Reising A, Yao C, Storey D, Webster TJ. *J Biomed Mater Res.* 2008; 87A:78–83.
98. Großner-Schreiber B, Herzog M, Hedderich J, Duck A, Hannig M, Griepentrog M. *Clin Oral Impl Res.* 2006; 17:736.
99. Ogawa T, Saruwatari L, Takeuchi K, Aita H, Ohno N. *J Dent Res.* 2008; 87:751. [PubMed: 18650547]
100. Hovgaard M, Chevallier J, Foss M, Besenbacher F. *Appl Phys Lett.* 2005; 87:073105.
101. Rautray TR, Narayanan R, Kwon T-Y, Kim K-H. *J Biomed Mater Res Part B: Applied Biomater.* 2010 in Press.
102. Hanawa T. *Mater Sci Eng A.* 1999; 267:260.
103. Hanawa T, Kamiura Y, Yamamoto S, Kohgo T, Amemiya A, Ukai H, Murakami K, Asaoka K. *J Biomed Mater Res.* 1997; 36:131. [PubMed: 9212398]
104. Saldana L, Barranco V, Gonzalez-Carrasco JL, Rodriguez M, Munuera L, Vilaboa N. *J Biomed Mater Res A.* 2007; 81:334. [PubMed: 17120220]
105. Saldana L, Vilaboa N, Valles G, Gonzalez-Cabrero J, Munuera L. *J Biomed Mater Res A.* 2005; 73:97. [PubMed: 15704115]
106. Engler AJ, Humbert PO, Wehrle-Haller B, Weaver VM. *Science.* 2009; 324:208–212. [PubMed: 19359578]
107. Kasemo B. *Surf Sci.* 2002; 500:656.
108. Anselme K. *Biomaterials.* 2000; 21:667. [PubMed: 10711964]
109. Puleo DA, Nanci A. *Biomaterials.* 1999; 20:2311. [PubMed: 10614937]
110. Kasemo B, Gold J. *Adv Dent Res.* 1999; 13:8. [PubMed: 11276751]
111. Hori N, Att W, Ueno T, Sato N, Yamada M, Saruwatari L, Suzuki T, Ogawa T. *J Dent Res.* 2009; 88:663. [PubMed: 19641155]
112. Nagassa ME, Daw AE, Rowe WG, Carley A, Thomas DW, Moseley R. *Clin Oral Implants Res.* 2008; 19:1317. [PubMed: 19040449]
113. Wilson CJ, Clegg RE, Leavesley DI, Percy MJ. *Tissue Eng.* 2005; 11:1. [PubMed: 15738657]
114. Gugutkov D, Altankov G, Hernández JCR, Pradas MM, Sánchez MS. *J Biomed Mater Res A.* 2010; 92:322. [PubMed: 19189388]
115. Sela MN, Badihi L, Rosen G, Steinberg D, Kohavi D. *Clin Oral Implants Res.* 2007; 18:630. [PubMed: 17484735]
116. Roach P, Farrar D, Perry CC. *J Am Chem Soc.* 2006; 128:3939. [PubMed: 16551101]
117. Hemmersam AG, Foss M, Chevallier J, Besenbacher F. *Colloids Surf.* 2005; 43:208.
118. Denis FA, Hanarp P, Sutherland DS, Gold J, Mustin C, Rouxhet PG, Dufrene YF. *Langmuir.* 2002; 18:819.
119. Dolatshahi-Pirouz A, Rechendorff K, Hovgaard MB, Foss M, Chevallier J, Besenbacher F. *Colloids Surf, B.* 2008; 66:53.
120. Hovgaard MB, Rechendorff K, Chevallier J, Foss M, Besenbacher F. *J Phys Chem B.* 2008; 112:8241. [PubMed: 18564871]
121. Rechendorff K, Hovgaard MB, Foss M, Zhdanov VP, Besenbacher F. *Langmuir.* 2006; 22:10885. [PubMed: 17154557]
122. Richert L, Variola F, Rosei F, Wuest JD, Nanci A. *Surf Sci.* 2010; 604:1445.
123. Cai K, Bossert J, Biointerfaces KDJSB. *Colloids Surf B Biointerfaces.* 2006; 49:136. [PubMed: 16621470]
124. Carpenter J, Khang D, Webster TJ. *Nanotechnology.* 2008; 19:505103. [PubMed: 19942760]

125. González-García C, Sousa SR, Moratal D, Rico P, Salmerón-Sánchez M. Colloids Surf B Biointerfaces. 2010 in Press.
126. Pollard TD, Earnshaw WC. Cell Biology. 2002
127. Ercan B, Webster TJ. Int J Nanomed. 2008; 3:477.
128. Chiang C-Y, Chiou S-H, Yang W-E, Hsu M-L, Yung M-C, Tsai M-L, Chen L-K, Huang H-H. Dental Mater. 2009; 25:1022.
129. Bauer S, Park J, Faltenbacher J, Berger S, Mark Kvd, Schmuki P. Integr Biol. 2009; 1:525.
130. Park J, Bauer S, Schmuki P, Mark Kvd. Nano Lett. 2009; 9:3157. [PubMed: 19653637]
131. Burns K, Yao C, Webster TJ. J Biomed Mater Res. 2009; 88A:561.
132. Puckett SD, Lee PP, Ciombor DM, Aaron RK, Webster TJ. Acta Biomater. 2010; 6:2352. [PubMed: 20005310]
133. Puckett SD, Taylor E, Raimondo T, Webster TJ. Biomaterials. 2010; 31:706. [PubMed: 19879645]
134. Price RL, Ellison K, Haberstroh KM, Webster TJ. J Biomed Mater Res A. 2004; 70:129. [PubMed: 15174117]
135. Oliveira, PTd; Nanci, A. Biomaterials. 2004; 25:403. [PubMed: 14585688]
136. Oliveira PTD, Zalzal SF, Beloti MM, Rosa AL, Nanci A. J Biomed Mater Res. 2007; 80A:554.
137. Gingell D. Symp Soc Exp Biol. 1993; 47:1. [PubMed: 8165560]
138. Sechler JL, Corbett SA, Wenk MB, Schwarzbauer JE. Ann NY Acad Sci. 1998; 857:143. [PubMed: 9917839]
139. Sepulveda JL, Gkretsi V, Wu C. Curr Top Dev Biol. 2005; 68:183. [PubMed: 16125000]
140. Mao Y, Schwarzbauer JE. Matrix Biol. 2005; 24:389. [PubMed: 16061370]
141. Mendonça G, Mendonça DB, Aragão FJ, Cooper LF. J Biomed Mater Res A. 2010; 94:169. [PubMed: 20128007]
142. Bajgai MP, Parajuli DC, Park S-J, Chu KH, Kang H-S, Kim HY. J Mater Sci: Mater Med. 2010; 21:685. [PubMed: 19851842]
143. Tosatti S, et al. J Biomed Mat Res A. 2004; 68:458.
144. Balasundaram G, Webster TJ. J Biomed Mat Res A. 2007; 80:602.
145. Secchi AG, et al. J Biomed Mat Res A. 2007; 83:577.
146. Auernheimer J, et al. BioChem. 2005; 6:2034.
147. Douglas T, Heinemann S, Hempel U, Mietrach C, Knieb C, Bierbaum S, Scharnweber D, Worch H. J Mater Sci: Mater Med. 2008; 19:1653. [PubMed: 17851735]
148. Hacking SA, Zuraw M, Harvey EJ, Tanzer M, Krygier JJ, Bobyn JD. J Biomed Mater Res A. 2007; 82:179. [PubMed: 17269149]
149. Takagi J, Petre BM, Walz T, Springer TA. Cell. 2002; 110:599. [PubMed: 12230977]
150. Alberts, B. Molecular biology of the cell. 5. 2008.
151. Wang, YL.; Discher, DE. Methods in Cell Biology - Cell Mechanics. Academic Press-Elsevier; Oxford, UK: 2007.
152. Biggs MJP, Richards RG, Dalby MJ. Nanomedicine: Nanotechnology, Biology, and Medicine. 2010 in Press.
153. Girard PP, Cavalcanti-Adam EA, Kemkemer R, Spatz JP. Soft Matter. 2007; 3:307.
154. Bustelo XR, Sauzeau V, Berenjano IM. BioEssays. 2007; 29:356. [PubMed: 17373658]
155. Watt FM, Jordan PW, O'Neill CH. Proc Natl Acad Sci USA. 1988; 85:5576. [PubMed: 2456572]
156. Adams JC, Watt FM. Development. 1993; 117:1183. [PubMed: 8404525]
157. Roskelley CD, Desprez PY, Bissell MJ. Proc Natl Acad Sci USA. 1994; 91:12378. [PubMed: 7528920]
158. Chen CS, Mrksich M, Huang S, Whitesides GM, Ingber DE. Science. 1997; 276:1425. [PubMed: 9162012]
159. Chen CS, Alonso JL, Ostuni E, Whitesides GM, Ingber DE. Biochem Biophys Res Commun. 2003; 307:355. [PubMed: 12859964]

160. McBeath R, Pirone DM, Nelson CM, Bhadriraju K, Chen CS. *Dev Cell*. 2004; 6:483. [PubMed: 15068789]
161. Albrektsson T, Wennerberg A. *Int J Prosthodont*. 2004; 17:544. [PubMed: 15543911]
162. Shalabi MM, Gortemaker A, Hof MAVt, Jansen JA, Creugers NH. *J Dent Res*. 2006; 85:496. [PubMed: 16723643]
163. Sul YT, Johansson C, Wennerberg A, Cho LR, Chang BS, Albrektsson T. *Int J Oral Maxillofac Implants*. 2005; 20:349. [PubMed: 15973946]
164. Davies JE. *J Dent Educ*. 2003; 67:932. [PubMed: 12959168]
165. Park JY, Davies JE. *Clin Oral Implants Res*. 2000; 11:530. [PubMed: 11168246]
166. Boyan BD, Lössdorfer S, Wang L, Zhao G, Lohmann CH, Cochran DL, Schwartz Z. *Eur Cell Mater*. 2003; 6:22. [PubMed: 14577052]
167. Kieswetter K, Schwartz Z, Hummert TW, Cochran DL, Simpson J, Dean DD, Boyan BD. *J Biomed Mater Res*. 1996; 32:55. [PubMed: 8864873]
168. Wilmosky, Cv; Bauer, S.; Lutz, R.; Meisel, M.; Neukam, FW.; Toyoshima, T., et al. *J Biomed Mater Res B Appl Biomater*. 2009; 89:165. [PubMed: 18780361]
169. Jäger M, Zilkens C, Zanger K, Krauspe R. *J Biomed Biotechnol*. 2007; 2007:69036. [PubMed: 18274618]
170. Guehennec LL, Soueidan A, Layrolle P, Amouriq Y. *Dental Mat*. 2007; 23:844.
171. Park KH, Heo SJ, Koak JY, Kim SK, Lee JB, Kim SH, Lim YJ. *J Oral Rehab*. 2007; 34:517.
172. Yang GL, FM FMH, Hu JA, Wang XX, Zhao SF. *Oral Surg Oral Med Oral Pathol Oral Radiol Endod*. 2009; 107:782. [PubMed: 19201624]
173. Meirelles L, Arvidsson A, Andersson M, Kjellin P, Albrektsson T, Wennerberg A. *J Biomed Mater Res A*. 2008; 87:299. [PubMed: 18181110]
174. Meirelles L, Melin L, Peltola T, Kjellin P, Kangasniemi I, Currie F, Andersson M, Albrektsson T, Wennerberg A. *Clin Implant Dent Relat Res*. 2008; 10:254.
175. Sul YT, Jönsson J, Yoon GS, Johansson C. *Clin Oral Implants Res*. 2009; 20:1146. [PubMed: 19719742]
176. Sul YT, Kang BS, Johansson B, Um HS, Park CJ, Albrektsson T. *J Biomed Mater Res A*. 2009; 89:942. [PubMed: 18470920]
177. Sul YT, Johansson CB, Byon E, Albrektsson T. *Biomaterials*. 2005; 26:6720. [PubMed: 15975649]
178. Sul YT, Johansson CB, Albrektsson T. *Int J Prosthodont*. 2006; 19:319. [PubMed: 16900812]
179. Tavares MG, Oliveira PTd, Nanci A, Hawthorne AC, Rosa AL, Xavier SPR. *Clin Oral Impl Res*. 2007; 18:452.
180. Guo J, Padilla RJ, Ambrose W, Kok IJD, Cooper LF. *Biomaterials*. 2007; 28:5418. [PubMed: 17868850]
181. Ellingsen JE, Johansson CB, Wennerberg A, Holmen A. *Int J Oral Maxillofac Implants*. 2004; 19:656.
182. Berglundh T, Abrahamsson I, Albouy JP, Lindhe J. *Clin Oral Implants Res*. 2007; 18:147. [PubMed: 17269959]
183. Abrahamsson I, Albouy JP, Berglundh T. *Clin Oral Implants Res*. 2008; 19:153. [PubMed: 18039334]
184. Stanford CM, Johnson GK, Fakhry A, Gratton D, Mellonig JT, Wanger W. *Appl Osseointegration Res*. 2006; 5:50.
185. Oxby G, Lindqvist J, Nilsson P. *Appl Osseointegration Res*. 2006; 5:68.
186. Mendes VC, Moineddin R, Davies JE. *Biomaterials*. 2007; 28:4748. [PubMed: 17697709]
187. Orsini G, Piattelli M, Scarano A, Petrone G, Kenealy J, Piattelli A, et al. *J Periodontol*. 2007; 78:209. [PubMed: 17274708]
188. Goené RJ, Testori T, Trisi P. *Int J Periodontics Restorative Dent*. 2007; 27:211. [PubMed: 17694944]
189. Vignoletti F, Sanctis Md, Berglundh T, Abrahamsson I, Sanz M. *J Clin Periodontol*. 2009; 36:688. [PubMed: 19563328]

190. Coelho PG, Cardaropoli G, Suzuki M, Lemons JE. J Biomed Mater Res B Appl Biomater. 2009; 88:387. [PubMed: 18395823]
191. Granato R, Marin C, Gil JN, Chuang SK, Dodson TB, Suzuki M, Coelho PG. Clin Implant Dent Relat Res. 2009 In press.
192. Park JW, Kim HK, Kim YJ, An CH, Hanawa T. Clin Oral Implants Res. 2009; 20:684. [PubMed: 19489932]
193. Park J-W, Suh J-Y, Chung H-J. J Biomed Mater Res A. 2008; 86:117. [PubMed: 17941022]
194. Suh J-Y, Jeung O-C, Choi B-J, Park J-W. Clin Oral Implants Res. 2007; 18:362.
195. Marin C, Granato R, Suzuki M, Gil JN, Piattelli A, Coelho PG. J Periodontol. 2008; 79:1942. [PubMed: 18834250]
196. Shibli JA, Grassi S, Piattelli A, Pecora GE, Ferrari DS, Onuma T, d'Avila S, Coelho PG, Barros R, Iezzi G. Clin Implant Dent Relat Res. 2009
197. Hetrick EM, Schoenfish MH. Chem Soc Rev. 2006; 35:780. [PubMed: 16936926]
198. Zhao L, Chu PK, Zhang Y, Wu Z. J Biomed Mater Res Part B: Appl Biomater. 2009; 91B:470. [PubMed: 19637369]
199. Lucke M, Schmidmaier G, Sadoni S, Wildemann B, Schiller R, Haas NP, Raschke M. Bone. 2003; 32:521. [PubMed: 12753868]
200. Hanawa T. J Artificial Organs. 2009; 12:73.
201. Puskas JE, Muñoz-Robledo LG, Hoerr RA, Foley J, Schmidt SP, Evancho-Chapman M, Dong J, Frethem C, Haugstad G. Nanomedicine and Nanobiotechnology. 2009; 1:451. [PubMed: 20049810]
202. Tsujino I, Ako J, Honda Y, Fitzgerald PJ. Expert Opinion on Drug Delivery. 2007; 4:287. [PubMed: 17489655]
203. Wieneke H, Dirsch O, Sawitowski T, et al. Catheter Cardiovasc Interv. 2003; 60:399. [PubMed: 14571494]
204. Grube, E. American College of Cardiology meeting; Chicago, USA. 2003.
205. Gottlow, J. The implant design and biological response. Wiley-Blackwell; Iowa: 2009.
206. Simmons CA, Meguid SA, Pilliar RM. J Biomed Mater Res. 2001; 55:63. [PubMed: 11426399]
207. Mouzin O, Soballe K, Bechtold JE. J Biomed Mater Res. 2001; 58:61. [PubMed: 11152999]
208. Simmons CA, Meguid SA, Pilliar RM. J Orthop Res. 2001; 19:187. [PubMed: 11347689]
209. Van OH, Duyck J, Vander SJ, Perre GVD, De SLCM, Puers R, Naert I. Clin Oral Implants Res. 1998; 9:407. [PubMed: 11429942]
210. De SE, Jaecques S, Vandamme K, Vander SJ, Naert I. Clin Oral Implants Res. 2005; 16:402. [PubMed: 16117763]
211. Duyck J, Slaets E, Sasaguri K, Vandamme K, Naert I. J Clin Periodontol. 2007; 34:998. [PubMed: 17935504]
212. Duyck J, Vandamme K, Geris L, Van OH, De CM, Vandersloten J, Puers R, Naert I. Arch Oral Biol. 2006; 51:1. [PubMed: 15922992]
213. Rubin CT, McLeod KJ. Clin Orthop Relat Res. 1994:165. [PubMed: 8118971]
214. Szmukler-Moncler S, Salama H, Reingewirtz Y, Dubruille JH. J Biomed Mater Res. 1998; 43:192. [PubMed: 9619438]
215. Vandamme K, Naert I, Geris L, Vander SJ, Puers R, Duyck J. J Clin Periodontol. 2007; 34:172. [PubMed: 17309592]
216. Brunski JB. Adv Dent Res. 1999; 13:99. [PubMed: 11276755]
217. Leucht P, Kim JB, Wazen R, Currey JA, Nanci A, Brunski JB, Helms JA. Bone. 2006; 40:919. [PubMed: 17175211]
218. Jokstad A, Braegger U, Brunski JB, Carr AB, Naert I, Wennerberg A. Int Dent J. 2003; 53:409. [PubMed: 14725421]
219. Lobo EG, Fang TD, Warren SM, Lindsey DP, Fong KD, Longaker MT, Carter DR. Bone. 2004; 34:336. [PubMed: 14962812]
220. Humphrey JD. J Biomech Eng. 2001; 123:638. [PubMed: 11783737]

221. Brunski, JB.; Currey, JA.; Helms, JA.; Leucht, P.; Nanci, A.; Wazen, RM. The healing bone-implant interface: role of micromotion and related strain levels in tissue. A. Jokstad; Iowa: 2008.
222. Claes L, Blakytyn R, Göckelmann M, Schoen M, Ignatius A, Willie B. J Orthop Res. 2009; 27:22. [PubMed: 18634011]
223. Claes LE, Wilke HJ, Augat P, Rübenacker S, Margevicius KJ. Clin Biomech (Bristol, Avon). 1995; 10:227.
224. Claes L, Schorlemmer PAPS, Konrads C, Ignatius A, Ehrnthaller C. J Orthop Res. 2008; 26:772. [PubMed: 18240329]
225. Albrektsson T, Wennerberg A. Int J Prosthodont. 2004; 17:536. [PubMed: 15543910]
226. Leven RM, Virdi AS, Sumner DR. J Biomed Mater Res A. 2004; 70:391. [PubMed: 15293312]
227. Masaki C, Schneider GB, Zaharias R, Seabold D, Stanford C. Clin Oral Implants Res. 2005; 16:650. [PubMed: 16307571]
228. Ogawa T, Nishimura I. J Dent Res. 2006; 85:566. [PubMed: 16723657]
229. Isa ZM, Schneider GB, Zaharias R, Seabold D, Stanford CM. Int J Maxillofac Implants. 2006; 21:203.
230. Discher DE, Mooney DJ, Zandstra PW. Science. 2009; 324:1673. [PubMed: 19556500]
231. Leucht P, Kim JB, Currey JA, Brunski JB, Helms JA. PLoS One. 2007; 2:e390-391-e390. [PubMed: 17460757]
232. Yim EK, Leong KW. Nanomedicine. 2005; 1:10. [PubMed: 17292053]
233. Dalby MJ, McCloy D, Robertson M, Agheli H, Sutherland D, Affrossman S, Oreffo RO. Biomaterials. 2006; 27:2980. [PubMed: 16443268]
234. Hansen JC, Lim JY, Xu LC, Siedlecki CA, Mauger DT, Donahue HJ. J Biomech. 2007; 40:2865. [PubMed: 17467715]
235. Sargeant A, Goswami T. Materials and Design. 2007; 28:155.
236. Li Y, Wong C, Xiong J, Hodgson P, Wen C. J Dent Res. 2010; 89:493. [PubMed: 20332331]
237. Donato TAG, Almeida LHd, Nogueira RA, Niemeyer TC, Grandini CR, Caram R, Schneider SG, Santos AR. Mat Sci Eng C-Bio S. 2009; 29:1365.
238. Lim JY, Dreiss AD, Zhou Z, Hansen JC, Siedlecki CA, Hengstebeck RW, Cheng J, Winograd N, Donahue HJ. Biomaterials. 2007; 28:1787. [PubMed: 17218005]
239. Das K, Bose S, Bandyopadhyay A. J Biomed Mater Res. 2009; 90A:225.
240. Noh H, Vogler EA. Biomaterials. 2007; 28:405. [PubMed: 17007920]
241. Alves CM, Reis RL, Hunt JA. J R Soc Interface. 2010; 7:1367. [PubMed: 20181561]
242. Anderson, et al. Semin Immunol. 2008; 20:86. [PubMed: 18162407]
243. Wazen RM, Tye CE, Goldberg HA, Hunter GK, Smith CE, Nanci A. J Histochem Cytochem. 2007; 55:35. [PubMed: 16957163]
244. Cochran DL. J Periodontol. 1999; 70:1523. [PubMed: 10632528]
245. Morton D, Jaffin R, Weber HP. Int J Oral Maxillofac Implants. 2004; 19(Suppl):103. [PubMed: 15635950]
246. Tolstunov L. Implant Dent. 2006; 15:341. [PubMed: 17172950]
247. Jaffin RA, Berman CL. J Periodontol. 1991; 62:2. [PubMed: 2002427]
248. Bain CA. Int J Oral Maxillofac Implants. 1996; 11:756. [PubMed: 8990637]
249. Fiorellini JP, Chen PK, Nevins M, Nevins ML. Int J Periodontics Restorative Dent. 2000; 20:366. [PubMed: 11203576]
250. Ostman P-O, Hellman M, Albrektsson T, Sennerby L. Clin Oral Impl Res. 2007; 18:409.
251. Albrektsson, T. Annual Meeting of the Academy of Dental Materials; Trieste (Italy). 2010. Editon edn
252. Svanborg LM, Andersson WAMM. J Biomed Mater Res B Appl Biomater. 2010; 92:462. [PubMed: 19957360]
253. Wennerberg A, Albrektsson T. Clin Oral Implants Res. 2009; 4:172. [PubMed: 19663964]
254. Marshall SJ, Bayne SC, Baier R, Toms AP, Marshall GW. Dent Mater. 2010; 26:e11. [PubMed: 20018362]

255. Popat KC, Eltgroth M, Desai TA. *Small*. 2007; 3:1878. [PubMed: 17935080]
256. Redey SA, Razzouk S, Rey C, Bernache-Assollant D, Leroy G, Nardin M, Cournot G. *J Biomed Mat Res*. 1999; 45:140.
257. Wei J, Yoshinari M, Takemoto S, Hattori M, Kawada E, Liu B, Oda Y. *J Biomed Mat Res B Appl Biomater*. 2007; 81:66.
258. Brandon S, Haimovich N, Yeger E, Marmur A. *J Colloid Interface Sci*. 2003; 263:237. [PubMed: 12804908]
259. Marmur A. *Soft Matter*. 2006; 2:12–17.
260. Meiron TS, Marmur A, Saguy IS. *J Colloid Interface Sci*. 2004; 274:637. [PubMed: 15144840]
261. Choudhary S, Haberstroh KM, Webster TJ. *Tissue Eng*. 2007; 13:1421. [PubMed: 17518735]
262. Fine E, Zhang L, Fenniri H, Webster TJ. *Int J Nanomedicine*. 2009; 4:91. [PubMed: 19516887]
263. Lewis G. *Journal of Biomedical Materials Research Part B: Applied Biomaterials*. 2008; 86B: 569.
264. Kukreja N, Onuma Y, Daemen J, Serruys PW. *Pharmacological Research*. 2008; 57:171. [PubMed: 18339557]
265. Yang L, Webster TJ. *Expert Opin Drug Deliv*. 2009; 6:851. [PubMed: 19637973]
266. Naguib N, Ye H, Gogotsi Y, Yazicioglu AG, Megaridis CM, Yoshimura M. *Nano Lett*. 2004; 4:2237.
267. Clair S, Variola F, Kondratenko M, Jedrzejowski P, Nanci A, Rosei F, Perepichka DF. *J Chem Phys*. 2008; 128:144705. [PubMed: 18412469]
268. Avila G, Misch K, Galindo-Moreno P, Wang HL. *Implant Dent*. 2009; 18:17. [PubMed: 19212234]
269. Palmieri A, Pezzetti F, Brunelli G, Martinelli M, Muzio LL, Scarano A, Degidi M, Piattelli A, Carinci F. *Implant Dent*. 2008; 17:100. [PubMed: 18332763]
270. Mizuno M, Fujisawa R, Kuboki Y. *J Cell Physiol*. 2000; 184:207. [PubMed: 10867645]
271. Reyes CD, Petrie TA, Burns KL, Schwartz Z, García AJ. *Biomaterials*. 2007; 28:3228. [PubMed: 17448533]
272. Dalby MJ, Gadegaard N, Tare R, Andar A, Riehle MO, Herzyk P, Wilkison CDW, Oreffo ROC. *Nat Mater*. 2008; 6:997. [PubMed: 17891143]
273. Hodgkinson T, Yuan XF, Bayat A. *Expert Rev Med Devices*. 2009; 6:621. [PubMed: 19911874]
274. Webster TJ, Ergun C, Doremus RH, Siegel RW, Bizios R. *J Biomed Mater Res*. 2000; 51:475. [PubMed: 10880091]

Biographies

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Figure 1. SEM images of vertically oriented TiO_2 nanotubes of different diameters. Scale bars: 200 nm. Adapted from Reference ⁴¹. Copyright 2009, Wiley.

Figure 2. Characteristic SEM images of Ti surfaces nanostructured by oxidative etching with different solutions (scale bar 100 nm). Reproduced with permission from Reference ⁵⁰. Copyright 2008, ACS.

Figure 3. SEM images of the structures formed on titanium metal at 250 °C, 5 h by varying the hexamine to NaOH molar ratio. Transition from nano-needles to nano-rods. Reproduced with permission from Reference ⁵⁷. Copyright 2010, IOP.

Figure 4.

Side-view SEM images of Ti surfaces after anodization at three different voltages and chemical removal of the alumina masks (above). Cytoskeletal and FA staining in hMSCs cultured on planar control and nanostructured Ti surfaces (below). Red, actin microfilaments; green, either tubulin microtubules; vimentin intermediate filaments or vinculin; blue, cell nucleus. Cells cultured on 15 nm high structured surfaces display a well-defined cytoskeleton and large focal adhesion sites, whereas on the higher structured surfaces cells have a less organized cytoskeleton and fewer, smaller focal adhesions. Picture adapted from Reference ⁴⁷. Copyright 2009, Elsevier.

Figure 5. Epifluorescence of calvarial osteogenic cells grown on control (A–C) and nano Ti (chemically oxidated with $\text{H}_2\text{SO}_4/\text{H}_2\text{O}_2$) (D–F) at 3 days, labeled with anti-fibronectin (A,D, green fluorescence) and anti-alfa 5 integrin (B,E, red fluorescence) antibodies. (C,F) Merged pictures of A,B and D,E, respectively, indicating that alfa 5 integrin expression colocalizes with extracellular fibronectin (yellowish). Integrin alfa-5 expression is up-regulated in osteogenic cells grown on nano Ti.⁵⁰ Blue fluorescence indicates cell nuclei (DAPI DNA stain). Scale bar = 20 μm , (unpublished picture).

Figure 6. Schematic drawing of focal adhesions (FAs) showing the interactions of Integrin molecules with other proteins on both sides of the lipid bilayer (unpublished picture).

Figure 7. SEM-Micrographs displaying 3T3-fibroblast adhering to micro-nanopatterns (A and B) and to an extended nanopattern (C). The dot spacing was 58 nm in all samples. Reproduced with permission from Reference ¹⁵³. Copyright 2007, RSC. Filopodia of primary calvarian cells sensing the nanotexture generated on titanium by oxidative nanopatterning (D–E)¹³⁵ (unpublished picture).

Figure 8. SEM micrograph of the CaP rodlike crystals. Reproduced with permission from Reference ¹⁷². Copyright 2009, Elsevier.

Figure 9.

Nanoporous surface (A) generated on miniature (1.0 mm inner diameter; 1.5 mm outer diameter; 2.4 mm head diameter; 0.4 mm head thickness; 1.6 mm implant length) Ti6Al4V screw-shaped implants (B) (Med Machining Inc, Miami Lakes, FL) by oxidative nanopatterning (unpublished picture).

Figure 10.

A nanoporous stent surface: a scanning electron microscopic view of a Setagon stent (x 20,000). Stent struts are gilded with this nanoporous layer (estimated porosity of 30%), which functions as a reservoir for prolonged release of antirestenotic drugs. Reproduced with permission from Reference ²⁰². Copyright 2007, Informa Healthcare.

Table 1

In vitro results with nanostructured metals.

Metal	Method	Nanoscale surface features	Tunable surface properties	Biological outcome	Ref.
Titanium, Ti6Al4V	Anodization	Nanotubes (diameter < 100 nm)	Pore size, nanotube length, oxide thickness, crystallinity	Enhanced osteogenic activity (adhesion, proliferation, bone matrix secretion and mineralization), promotion of focal adhesions assembly, stimulation of hMSC cell activity (growth and differentiation), enhanced adhesion of chondrocytes and keratinocytes, limitation of bacterial growth	36, 38–41, 62, 127, 128, 131–133, 239
Titanium, Ti6Al4V, CrCoMo, Tantalum	Oxidative nanopatterning	Nanoporous structure (20–100 nm diameter)	Micro and nanotopographies, oxide thickness, chemical inclusions (F, S)	Differential activity (enhancement of osteoblastic activity and limitation of fibroblast growth), gene and protein (BSP, OPN, Alkaline phosphatase, Runx2) upregulation, stimulation of HUC stem cells, antibacterial effect	50–53, 135, 136, 141
Titanium	Hydrotherma l+NaOH treatment	Nanoscaffolds, nanoflowers, nanoleaves, nanoneedles, nanorods, nanotubes and multi-scale octahedral whiskers	Dimensions of nanostructures	Enhanced protein adsorption	57
Titanium	Sol-gel and Chemical vapor deposition (CVD)	Nanotopographical niobium oxide and diamond-like carbon	Nanoscale roughness, layer thickness, crystallinity	Enhanced osteoblast adhesion and proliferation	58, 59, 61, 63, 142
Titanium, Ti6Al4V	Biochemical functionalization	Molecular coatings (molecules with RGD motifs and ECM proteins, antibacterial agents, antibiotics, nanorosette, non-fouling and anticoagulant agents)	Molecular functional groups, coating thickness	Enhanced osteoblast attachment, adhesion, proliferation, gene expression, antifouling and anticoagulant properties	38, 64, 67, 68, 77–84, 90–92, 143, 145, 146
Titanium, Ti6Al4V	Plasma deposition, physical and e-beam vapor deposition (PVD), electron-gun evaporation	Nanostructured titanium, titanium nitride, zirconium nitride, tantalum coatings	Nanoscale roughness, coating thickness	Enhanced osteoblast proliferation and Ca deposition, enhanced fibroblast attachment	97–100
Titanium	Ion implantation	Implanted Ca ²⁺ , F ⁻ , Na ⁺ , P ⁺ , etc	Concentration and depth distribution of implanted elements	Improved osteoconductivity	101–103
Titanium, Ti6Al4V	Thermal oxidation/Annealing and DC reactive magnetron sputtering	Crystalline TiO ₂ layers	Nanotopography, crystalline structure	Improved osteoblast activity, enhanced osteoblast adhesion, spreading, proliferation and differentiation (on anatase)	45, 104, 105

Table 2

In vivo results with nanostructured metals.

Metal/Implant	Method	Nanoscale surface features	Surface chemistry (Doping)	Surgical protocol	Biological outcome	Ref.
Titanium rod	Anodization	TiO ₂ nanotubes (30 nm diameter)	N/A	Pig frontal skull - 3, 7, 14, 30, 90 days	Stimulation of collagen-I expression	168
Titanium screw, rod	Electrochemical deposition, Promimic HA ^{Nano®}	HA nanocrystals	Ca, P	Rabbit proximal tibia - 4, 6, 12 weeks;	Improved bone- to-implant contact, early bone formation	172, 173
Titanium screw	Micro-arc oxidation (MAO)	N/A	Mg	Rabbit Tibia - 3, 6 weeks	Improved bone- integration and removal torque	175-178
Titanium screw	Oxidative nanopatterning	Nanopores (20-25 nm diameter)	N/A	Dog mandible - 3, 8 weeks	Improved bone- to-implant contact	179
Titanium rod, screw	TiO ₂ blasting+HF acid treatment	Uncharacterized features (50-200 nm in size)	F	Rat tibia - 21 days; Dog mandible - 2 weeks	Stimulation of osteoblastic gene expression, enhanced bone formation and bone-implant fixation, good osteointegration and stability after 1 year	55, 180-183, 185
Titanium rectangular plates, rod,	Dual acid etching (DAE) +Discrete crystalline deposition (DCD)	CaP nanocrystals (20-100 nm in size) and nanoparticles	N/A	Rat femur - 9 days; Human posterior maxilla - 4, 8 weeks, 2 months; Dog tooth socket - 4 h, 1, 2, 4 and 8 weeks	Improved bone- to-implant contact and bone formation, enhanced tissue mineralization, less bone desorption	186-189
Titanium rod	Ion beam assisted deposition (IBAD)	Nano-thick (20-500 nm) bioactive (Ca- and P-based) ceramic layers	Ca, P	Dog proximal tibia - 2, 4 weeks	Improved fixation	190, 191
Titanium screw	Hydrothermal alkaline treatment	Uncharacterized features (about 100 nm in size)	Ca	Rabbit femur -6 weeks;	Improved bone- to-implant contact and removal torque, enhanced osteoconductivity	171, 192-194
Titanium screw	Bioceramic grit-blasting (BGB) +Acid Etching (AA)	Nanorough topography	Ca, P	Dog tibia - 2, 4 weeks, 2 months	Early biomechanical fixation and improved bone- to-implant contact	195, 196